ABSTRACT

Owing to the significance of semiconducting transition metal oxides, doped with transition metals, in the area of spintronics, we prepared zinc oxide co-doped with nickel and magnesium, following chemical co-precipitation strategy. Since our focus was also to look at the effect of the amount of the co-dopants on the bandwidth of zinc oxide, we prepared six different samples of zinc oxide co-doped with nickel and magnesium, varying the amount of the co-dopants. The prepared samples were characterized following Energy Dispersive X-ray Analysis, powder X-ray Diffraction Analysis and solid-state UV-visible spectral analysis. The Tau plots were utilized for the determination of the bandwidth of all the prepared samples. The bandwidth of all the prepared samples thus obtained was compared with the bandwidth of pure undoped zinc oxide. For this, we also prepared pure undoped zinc oxide following a chemical precipitation strategy. It was observed that the bandwidth of all the prepared samples, zinc oxide co-doped with nickel and magnesium, is greater than that of the pure undoped zinc oxide that we prepared. Again, we observed that the bandwidth of pure zinc oxide increases as the amount of nickel and magnesium in the crystal lattice of zinc oxide increases.

Keywords: Zinc Oxide, Co-doped Metal Oxides, Bandwidth, Tau Plot, Energy Dispersive X-ray Analysis

INTRODUCTION

The area, electronics is exclusively based on the manipulation of the negative charge of the electron, and here the spin of the electron is not taken into account. However, in spintronics, the spin or the spin and the charge together are considered, and the spin of the electron is manipulated for carrying the information. To find application in the field of spintronics, a material is supposed to exhibit ferromagnetism at room temperature. In this scenario, the so-called half-metallic oxides are of greater significance.\(^1,2\) Again, the so-called dilute magnetic semiconductors are also suitable for spintronics. These are generated by doping the semiconductors with magnetic impurities.\(^3,4\) By doping with magnetic impurities, it is possible to alter the bandwidth and ferromagnetic properties of these semiconductors. The bandwidth engineering and tuning of ferromagnetic properties in the oxides of transition metals are possible without doping with magnetic impurities. It is achieved by doping these oxides with various transition metals. A large number of reports are available on the bandwidth engineering and tuning of room temperature ferromagnetic properties of the transition metal oxide, ZnO, where the bandwidth and ferromagnetic properties of this oxide are altered doping with other d-block metals or f-block metals.\(^5,6\) There are reports from our group on the bandwidth engineering of ZnO and studies on the photocatalytic properties of the doped ZnO.\(^7,8,9\) A large volume of work in this area is carried out with ZnO because of its very interesting properties and easy tunability of band width.\(^10,11,12\)
EXPERIMENTAL

Chemicals and Reagents Used
All the chemicals and reagents used are of either Laboratory Reagent (LR) grade or Analytical Reagent (AR) grade. They were used as such without further purification. The various reagents used in the present work include ZnSO$_4$·7H$_2$O (NICE, LR grade), CaSO$_4$·2H$_2$O (NICE, LR grade), MgSO$_4$·7H$_2$O (NICE, AR grade), NiSO$_4$·6H$_2$O (NICE, LR grade) and NaOH (MERK). For the preparation of various solutions and the washing of the final products, demineralized water was used.

Preparation of Pure ZnO
To a solution of 11.506 g of ZnSO$_4$·7H$_2$O in 100 ml demineralized water, kept at constant stirring over a magnetic stirrer at room temperature, a solution of 3.296 g of NaOH in 50 ml demineralized water was added slowly in a thin stream. It was kept for 15 minutes in the stirring condition. The precipitated Zn(OH)$_2$ was filtered through a sintered glass crucible, G4. The precipitate was washed with demineralized water till the washings were free from Zn$^{2+}$ ions, SO$_4^{2-}$ ions and unreacted NaOH. The presence of NaOH in the washings was checked with a solution of phenolphthalein (no pink or pale pink color). A solution of BaCl$_2$ was used to confirm that the washings are free from SO$_4^{2-}$ ions (no white precipitate of BaSO$_4$ or milky appearance). When the washings, after acidification with dilute H$_2$SO$_4$, were treated with a drop of cobalt nitrate solution and ammonium mercuric thiocyanate reagent, no pale-blue precipitate or color was developed indicating the absence of Zn$^{2+}$ ions in the washings. The precipitate was finally washed with acetone and dried in an air oven at 50 °C. Thermal decomposition of Zn(OH)$_2$ at 100 °C - 250 °C yields ZnO. In this work, the dried precipitate of Zn(OH)$_2$ was taken in a silica crucible and was heated in a muffle furnace to 150 °C from room temperature and kept at 150 °C for 2 hours. After 2 hours, the furnace was switched off and allowed to cool along with the sample to room temperature. The ZnO thus obtained was powdered and used for further study.

Preparation of ZnO Co-doped with Ni and Mg
To a solution of 2.875 g of ZnSO$_4$·7H$_2$O, 0.2628 g of NiSO$_4$·6H$_2$O and 0.7394 g of MgSO$_4$·7H$_2$O in 100 ml demineralized water, kept at constant stirring over a magnetic stirrer at room temperature, a solution of 0.8 g of NaOH in 50 ml demineralized water was added slowly in a thin stream. It was kept at the constant stirring condition for 15 minutes. The precipitated Zn(OH)$_2$ co-doped with Ni(OH)$_2$ and Mg(OH)$_2$ was filtered through a sintered glass crucible, G4. The precipitate was washed with demineralized water till the washings were free from Zn$^{2+}$ ions, SO$_4^{2-}$ ions, Ni$^{2+}$ ions, Mg$^{2+}$ ions and unreacted NaOH. The washings were checked for the presence or absence of the species Zn$^{2+}$ ions, SO$_4^{2-}$ ions and unreacted NaOH as explained under preparation of pure ZnO. When the washings were treated with a few drops of dimethyl glyoxime reagent, after adding a few drops of NH$_4$Cl solution and ammoniacal oxime reagent when no pale-yellow precipitate or color was formed indicating the absence of Zn$^{2+}$ ions, the washings were filtered through a sintered glass crucible, G4. The precipitate was washed with demineralized water till the washings were free from Zn$^{2+}$ ions, SO$_4^{2-}$ ions, Ni$^{2+}$ ions, Mg$^{2+}$ ions and unreacted NaOH. The washings were finally washed with acetone and dried in an air oven at 50 °C. Thermal decomposition of Zn(OH)$_2$ at 100 °C - 250 °C yields ZnO, Ni(OH)$_2$ at 230 °C - 360 °C yields NiO and Mg(OH)$_2$ at 360 °C gives MgO. In this work, the dried precipitate of Zn(OH)$_2$ co-doped with Ni(OH)$_2$ and Mg(OH)$_2$ was taken in a silica crucible and was heated in a muffle furnace to 450 °C from room temperature and kept at 450 °C for 6 hours. After 6 hours, the furnace was switched off and allowed to cool along with the sample to room temperature. Keeping the amount of ZnSO$_4$·7H$_2$O at 2.875 g and that of NiSO$_4$·6H$_2$O at 0.2628 g in the feed, three more samples of ZnO co-doped with Ni and Mg were prepared, following the same procedure, taking 0.4929 g, 0.2464 g and 0.9859 g of MgSO$_4$·7H$_2$O, separately. Here the amount of MgSO$_4$·7H$_2$O in the feed was varied. Similarly, keeping the amount of ZnSO$_4$·7H$_2$O at 2.875 g and that of MgSO$_4$·7H$_2$O at 0.2464 g in the feed, two more samples of ZnO co-doped with Ni and Mg were prepared to take 0.7885 g and 0.5257 g of NiSO$_4$·6H$_2$O, separately. Here the amount of NiSO$_4$·6H$_2$O in the feed was varied. A total of six samples of ZnO co-doped with Ni and Mg (samples A, B, C, D, E, and F) were prepared, as summarised in Table-1. All co-doped samples thus obtained were powdered and used for further study.
Table-1: Amount of ZnSO₄.7H₂O, NiSO₄.6H₂O and MgSO₄.7H₂O in the Feed

<table>
<thead>
<tr>
<th>Sample Name</th>
<th>Amount of ZnSO₄.7H₂O in gram in the feed</th>
<th>Amount of NiSO₄.6H₂O in gram in the feed</th>
<th>Amount of MgSO₄.7H₂O in gram in the feed</th>
</tr>
</thead>
<tbody>
<tr>
<td>C</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.2464</td>
</tr>
<tr>
<td>B</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.4929</td>
</tr>
<tr>
<td>A</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.7394</td>
</tr>
<tr>
<td>F</td>
<td>2.875</td>
<td>0.5257</td>
<td>0.9859</td>
</tr>
<tr>
<td>E</td>
<td>2.875</td>
<td>0.9859</td>
<td>0.2464</td>
</tr>
<tr>
<td>D</td>
<td>2.875</td>
<td>0.7885</td>
<td>0.2464</td>
</tr>
</tbody>
</table>

Characterization of the Samples

Energy Dispersive X-ray Analysis (EDAX)
The elements present in the prepared samples were identified by Energy Dispersive X-ray Analysis, EDAX, using the instrument CARL ZEISS EVO 18 Secondary Electron Microscope with EDS. The elemental analysis was carried out with sample C.

X-Ray Diffraction (XRD) Analysis
For the characterization of the crystal structure of the pure undoped ZnO and co-doped ZnO, powder X-ray diffraction (powder XRD) analysis of the prepared samples was carried out on a BRUKER D8 Advance XRD instrument.

UV-visible Spectral Analysis
To determine the bandwidth of the prepared samples, the solid-state UV-visible spectral analysis of the samples was performed on a Shimadzu UV 2600 spectrophotometer.

RESULTS AND DISCUSSION

Elemental Analysis
The elemental analysis was carried out with sample C. In sample C, the expected elements include zinc (Zn), nickel (Ni), magnesium (Mg) and oxygen (O). The result of EDAX for sample C is presented in Fig.-1. The result confirms the presence of these elements in the prepared sample C. In addition to these expected elements, the sample also contains sulfur (S). The presence of S in the co-doped sample is confirmed by the peak between 2 and 3 on the abscissa in Fig.-1.

![Fig.-1: EDAX of Sample C](image)

XRD Analysis
The result of the slow scan powder XRD analysis of the prepared samples are presented in Fig.-3 and Fig.-4 along with the XRD analysis result of the pure undoped ZnO, Fig.-2. The powder XRD patterns of the co-doped samples were recorded in the 2θ range of 10° - 80°. In all the x-ray diffractograms, prominent XRD lines are observed at 2θ of 31.45, 34.14, 36.04, 47.29, 56.23, 62.58, 66.98, 67.87 and 70.16°.
These lines could be indexed, respectively, as (100), (002), (101), (102), (110), (103), (200), (112) and (201). They correspond to the hexagonal wurtzite phase of ZnO [JCPDS 36-1451]. Therefore, undoped and co-doped samples possess hexagonal ZnO phase.

Furthermore, the peaks below 30° at the 2θ angle of 18.7, 20, 21.07, 22.38, 24, 25.06 etc. are perfectly in match with Zn₃O(SO₄)₂ in the prepared co-doped samples (JCPDS card number 000321475). The presence of S in the co-doped sample is confirmed by the peak between 2 and 3 on the abscissa in the EDAX, Fig.-1.

The 2θ corresponding to (101) plane of the co-doped samples is summarised and presented in Table-2. The 2θ corresponding to (101) plane of pure undoped ZnO that we prepared is 36.45°. Concerning this value, all the prepared co-doped samples showed a shift in the peak towards the lower angle, revealing that the dopants Ni and Mg are well incorporated in the crystal lattice of ZnO.

<table>
<thead>
<tr>
<th>Sample Name</th>
<th>Amount of ZnSO₄·7H₂O in gram in the feed</th>
<th>Amount of NiSO₄·6H₂O in gram in the feed</th>
<th>Amount of MgSO₄·7H₂O in gram in the feed</th>
<th>2θ in degree corresponding to (101) plane</th>
</tr>
</thead>
<tbody>
<tr>
<td>C</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.2464</td>
<td>35.96</td>
</tr>
<tr>
<td>B</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.4929</td>
<td>36.06</td>
</tr>
<tr>
<td>A</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.7394</td>
<td>35.95</td>
</tr>
<tr>
<td>F</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.9859</td>
<td>36.01</td>
</tr>
<tr>
<td>E</td>
<td>2.875</td>
<td>0.5257</td>
<td>0.2464</td>
<td>36.04</td>
</tr>
<tr>
<td>D</td>
<td>2.875</td>
<td>0.7885</td>
<td>0.2464</td>
<td>36.09</td>
</tr>
</tbody>
</table>

The 2θ corresponding to (101) plane of the co-doped samples is summarised and presented in Table-2. The 2θ corresponding to (101) plane of pure undoped ZnO that we prepared is 36.45°. Concerning this value, all the prepared co-doped samples showed a shift in the peak towards the lower angle, revealing that the dopants Ni and Mg are well incorporated in the crystal lattice of ZnO.
There are reports that the lattice distortion due to the defects (vacancies, interstitials, substitutions, local structure transformations etc.) may cause the shift in the position of the XRD peak. Based on the type of strain, tensile or compressive, in the crystal, the peak position shifts, respectively, towards higher or lower angle. The shifting of the peak position to a lower angle with an increase or decrease of doping amount may indicate the expansion of the lattice parameters. The shift in the peak in the prepared co-doped samples towards lower angles further confirms the incorporation of the dopants, Ni and Mg, in ZnO crystal lattice.

**Calculation of Bandwidth**

The UV-visible spectral data of the co-doped ZnO were used for the bandwidth determination. Theoretical bandwidth energy can be evaluated using Tauc’s relation. This relation is given by \( \alpha h \nu = D (h \nu - E)^n \). The value of \( n \) depends on the type of transition. That is, whether the transition is direct or indirect allowed transition or direct or indirect forbidden transition. In the present work, the value of \( n \) is equal to \( \frac{1}{2} \) since in ZnO the transitions involved are direct allowed transitions. Now, the Tauc’s relation for ZnO takes the form \( (\alpha h \nu)^2 = D^2 (h \nu - E) \). Based on this relation, it is possible to have a plot of \( (\alpha h \nu)^2 \) on ordinate vs. \( (h \nu) \) on the abscissa, the Tau plot. When the plot is extrapolated to abscissa, it will meet with a point on the abscissa; at this point \( (\alpha h \nu)^2 \) is equal to zero. The value of \( (h \nu) \), usually in eV, at this point on abscissa, will be equal to the bandwidth energy. Here in this work, the determination of bandwidth energy of all the prepared co-doped samples was carried out constructing Tau plot. The Tau plots are presented in Figures-5 to 10 for the co-doped samples, along with the Tau plot for pure undoped ZnO, Fig.-11. The bandwidth of the pure undoped ZnO that we prepared is 3.19 eV at room temperature. This value is less than the value available in the literature being 3.37 eV at room temperature. Thus, the actual bandwidth energy depends on the way of preparing the sample. The result of bandwidth calculation for the prepared co-doped samples is summarised and presented in Table-3.
From Table-3, it is clear that the bandwidth of the samples is greater than that of the pure undoped ZnO that we prepared, for this, it is 3.19 eV. Thus, the bandwidth of ZnO increases on co-doping with Ni and Mg.

Table-3: Bandwidth of the Co-doped Samples

<table>
<thead>
<tr>
<th>Sample Name</th>
<th>Amount of ZnSO$_4$.7H$_2$O in gram in the feed</th>
<th>Amount of NiSO$_4$.6H$_2$O in gram in the feed</th>
<th>Amount of MgSO$_4$.7H$_2$O in gram in the feed</th>
<th>Bandwidth in eV</th>
</tr>
</thead>
<tbody>
<tr>
<td>C</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.2464</td>
<td>3.265</td>
</tr>
<tr>
<td>B</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.4929</td>
<td>3.27</td>
</tr>
<tr>
<td>A</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.7394</td>
<td>3.28</td>
</tr>
<tr>
<td>F</td>
<td>2.875</td>
<td>0.2628</td>
<td>0.9859</td>
<td>3.29</td>
</tr>
<tr>
<td>E</td>
<td>2.875</td>
<td>0.5257</td>
<td>0.2464</td>
<td>3.27</td>
</tr>
<tr>
<td>D</td>
<td>2.875</td>
<td>0.7885</td>
<td>0.2464</td>
<td>3.28</td>
</tr>
</tbody>
</table>

Table-3 shows that for samples C, B, A and F, the bandwidth energy increases as the amount of MgSO$_4$.7H$_2$O in the feed increases keeping the amount of NiSO$_4$.6H$_2$O fixed. Thus, it is clear that the bandwidth of ZnO increases as the amount of the dopant, Mg increases.

For samples E and D, the bandwidth increases as the amount of NiSO$_4$.6H$_2$O in the feed increases keeping the amount of MgSO$_4$.7H$_2$O fixed. This again shows that the bandwidth of ZnO increases as the amount of the dopant, Ni increases in the crystal lattice of ZnO. Thus, it can be very well concluded that the bandwidth of pure ZnO increases as the amount of Ni and Mg in the crystal lattice of ZnO increases. It is interesting to note from Table-3 that the bandwidth of samples B and E is the same even though the amount of NiSO$_4$.6H$_2$O and the amount of MgSO$_4$.7H$_2$O in the feed are different. That is, even though the amount of Ni and Mg in the crystal lattice of ZnO, in these two samples, is different. Similar is the case with co-doped samples A and D.
Ni and Mg co-doped ZnO were prepared following a chemical co-precipitation strategy. The co-doped samples were prepared to vary the amount of Ni and Mg. Pure undoped ZnO was also prepared to adopt a chemical precipitation strategy. The co-doped samples were characterized by EDAX, powder XRD and UV-visible analysis. The EDAX revealed the presence of S in the crystal lattice of co-doped ZnO. The XRD supports this observation revealing the presence of the phase $\text{Zn}_3\text{O(SO}_4\text{)}_2$ in the co-doped samples. The $2\theta$ corresponding to (101) plane of pure undoped ZnO that we prepared is 36.45°. Concerning this value, all the prepared co-doped samples showed a shift in the peak towards the lower angle, revealing that the dopants Ni and Mg are well incorporated in the crystal lattice of ZnO. On co-doping, the bandwidth of ZnO increases concerning that of pure undoped ZnO that we prepared. The bandwidth of the prepared co-doped samples was found to increase as the amount of the co-dopants increases.

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