

GROWTH AND CHARACTERIZATION OF PROMISING NLO CRYSTAL: META NITROANILINE

K. Kumar^{1*}, K. Selvaraju¹, P. Baskaran¹, N. Senthilvelan² and G. Rajarajan^{3,4}

¹PG and Research Department of Physics, Government Arts College, Ariyalur-621 113, India

²Department of Physics, Arignar Anna Government Arts College, Attur-636 121, India

³Department of Physics, Vidhya Mandhir Institute of Technology, Erode-638 052, India

⁴VMIT Centre for Renewable Energy and Materials Science, Vidhya Mandhir Institute of Technology, Erode-638 052, India

*E-mail: kumar.physics2011@gmail.com

ABSTRACT

A good optical quality Meta Nitroaniline (mNA) single crystal was grown by slow evaporation method at room temperature. The grown crystals have been subjected to single crystal X-ray diffraction (XRD) to determine the unit cell dimensions. The Fourier Transform Infrared (FTIR) spectra have been recorded in the range 4000-400 cm⁻¹. UV-Vis spectrum shows that the grown crystals have wide optical transparency in the entire visible region. The optical energy band gap of the grown crystal is found to be 4.3 eV. This high value of band gap energy and the lower cut-off wavelength of the crystal assert the suitability of the crystal for photonic and optical applications. The grown crystals were subjected to other characterizations such as microhardness, thermal and NLO studies and the results are discussed in detail.

Keywords: single crystals, optical characterization, thermal study, elastic stiffness constant, XRD, FTIR, mNA.

© RASAYAN. All rights reserved

INTRODUCTION

These days, the nonlinear optics (NLO) field is getting a lot of attainment in science and technology since its progressions afford the foundation for the development of frequency up-converters and optical switches¹⁻³. Nonlinear optical materials capable of efficient second harmonic generation have been actively sought over the last three decades due to commercial importance. For this reason, scientists are constantly carrying out investigations towards the development of high-performance NLO materials with acceptable structural integrity in the solid-state. Hence, there is a special interest in exploring organic molecules in view of their ability to act as building blocks for the design of a widespread amount of molecular architectures⁴⁻⁷. Furthermore, the relatively good hyperpolarizability (β) values of some of these organic molecules make them even more attractive candidates for the development of highly efficient NLO materials^{8, 9}. Hence recent search is concentrated on organic materials due to their large nonlinearity, high resistance to laser-induced damage, low angular sensitivity and good mechanical hardness.

Among organic compounds, nitroaniline derivatives have displayed large β values due to the asymmetric charge distribution induced by the attachment of the electron-donating (-NH₂) and the electron-withdrawing (-NO₂) groups to the phenyl ring¹⁰⁻¹². For instance, nitroaniline is a good example of a push-pull system due to the degree of conjugation induced by the location of the substituents -NH₂ and -NO₂ groups at para position to each other, which enhances its β value¹³. However, most of these compounds exhibit large molecular dipole moments, which leads to antiparallel molecular arrangements and consequently centrosymmetric crystal structures¹⁴. It is well known that the hydrogen bond plays an important role in controlling the molecular assemblies due to its directional characteristics¹⁵. For this reason, strong hydrogen bonds (N-H/O, OH/O) are the most important elements in crystal engineering. Though, in the lack of these conventional hydrogen bonds, weaker interactions are answerable for crystal packing¹⁶.

In the instance of nitroaniline derivatives, it was found that they usually associate through intermolecular hydrogen bonds between the $-NH_2$ and $-NO_2$ groups giving the development of infinite polar chains, which can be useful for the design of efficient NLO materials¹⁷. Here in the present study due to the lack of work spotted in the literature encouraged us to make the above-mentioned studies on Meta Nitroaniline (mNA) crystals were grown and subjected to various characterizations and results are reported.

EXPERIMENTAL

The commercially existing Meta Nitroaniline (mNA) is purified by repetitive recrystallization procedure. The repeated recrystallized materials have been used for growth as charge material. Since Meta Nitroaniline (mNA) is soluble in Methanol and it is used as a solvent for crystal growth. The saturated solution of Meta Nitroaniline was obtained by dissolving the charge material into the solvent with continuous stirring of the solution using a magnetic stirrer at room temperature (305 K).

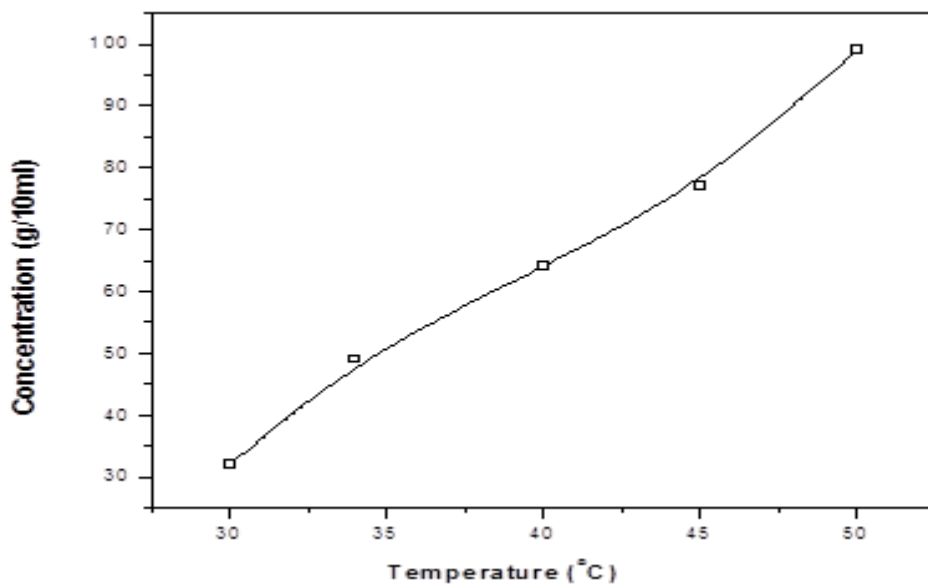


Fig.-1: Solubility Curve of mNA

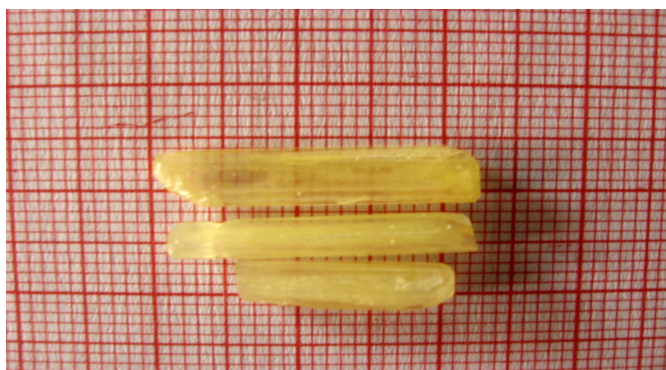


Fig.-2: Grown Crystal of mNA

On reaching saturation, the equilibrium concentration of the solute was determined by gravimetric analysis. The beaker containing the solution was optimally closed for controlled evaporation. Transparent yellow single crystals were obtained from the mother solution after a period of 15 days. Figure-1 shows the Solubility curves of mNA. Figure-2 shows the as-grown crystal of mNA with an optimized solution pH value of 3.7.

RESULTS AND DISCUSSION

Powder X-Ray Diffractions Studies

The structural properties of single crystals of mNA have been studied by X-Ray Powder diffraction technique. The X-Ray diffraction studies were carried out using SEIFERT diffractometer with $\text{CuK}\alpha_1$ ($\lambda = 1.5406 \text{ \AA}$) radiation. The powdered samples were scanned over the range of 10° - 50° at a rate of 1° per minute. From the powdered X-Ray data, various planes of reflections were indexed using XRDA 3.1 program and the lattice parameters were evaluated and agree with reported values¹⁸. The Powder X-ray diffraction pattern is shown in Figure 3. The crystallographic data are given in Table-1. Figure-4 represents the molecular packing of the mNA in the unit cell.

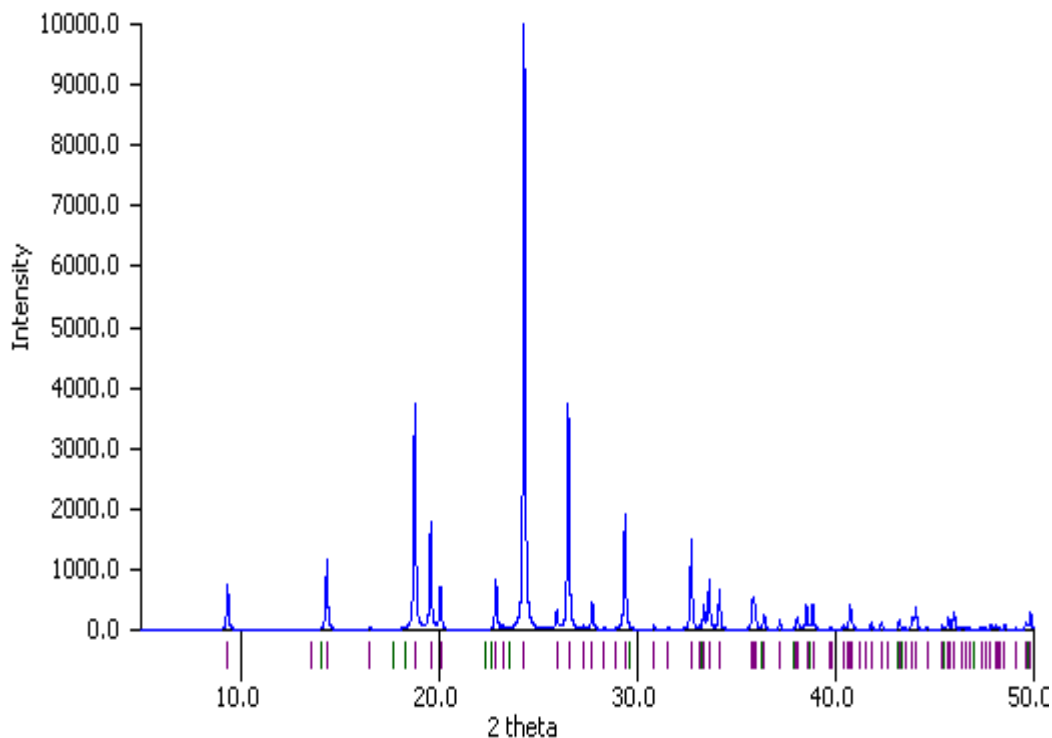


Fig.-3: The XRD pattern of mNA

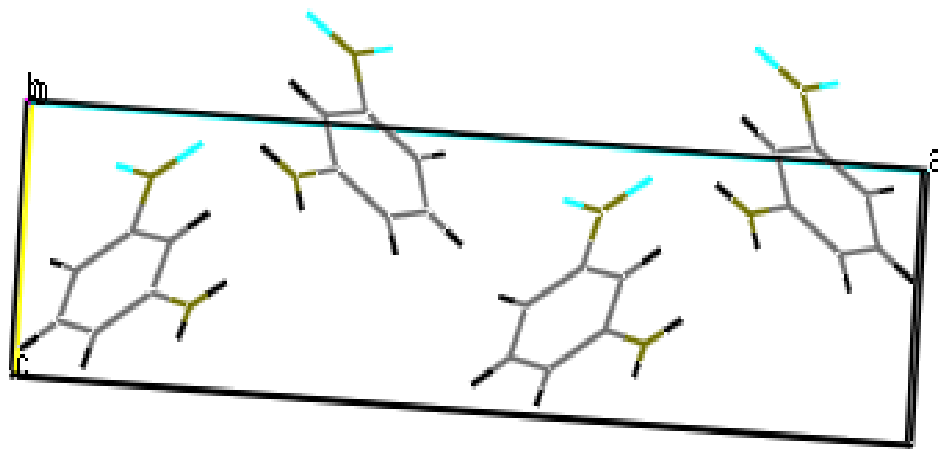


Fig.-4: Molecular packing of the mNA in the unit cell

Table-1: Crystallographic Data of mNA

Identification code	mNA
Empirical formula	C ₆ H ₆ N ₂ O ₂
Formula weight	138.1247
Crystal structure	Orthorhombic
Space group	Pca2(1)
Cell parameters	
a(Å)	18.873
b(Å)	6.5212
c(Å)	4.9980
α(°)	90.00
β(°)	90.00
γ(°)	90.00
Volume (Å) ³	615.13
Z	4

UV-Visible Transmittance Study

The UV-Vis transmission spectrum of pure and mNA crystals was recorded in the range of 200-1100 nm and is shown in the Fig.-5. The instrument used was LAMBDA-35 UV-Vis spectrophotometer. From the spectrum, it is seen that the lower cut off wavelength of mNA was around 280 nm. The spectrum further indicates that the crystal has a wide optical window from 280 nm to 1100 nm.

The transparency in the entire visible region shows that this material is a best suitable candidate for optoelectronic application^{19, 20}.

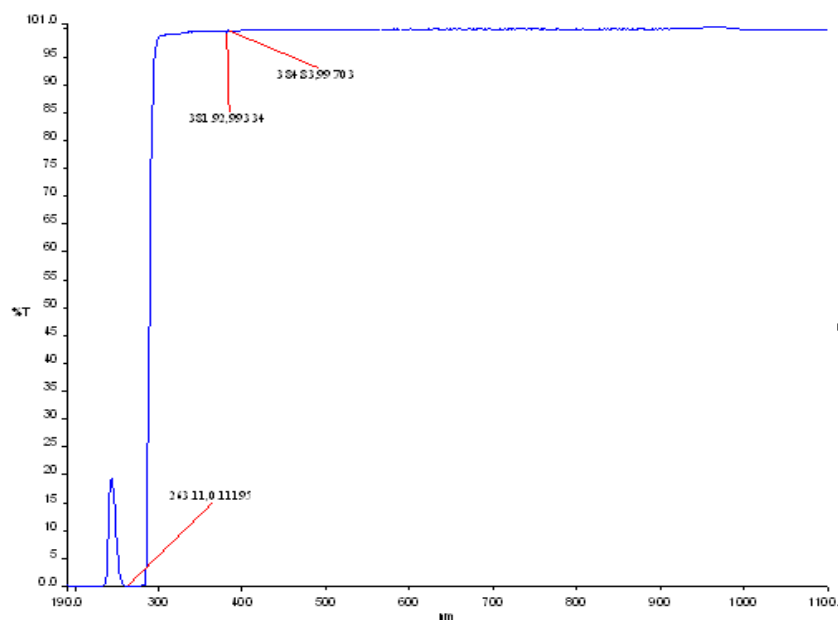


Fig.-5: UV-Vis Spectrum of mNA crystal

Energy Band Gap

The variant of $(\alpha h\nu)^2$ versus incident photon energy ($h\nu$) by means of the Tauc relation in the fundamental absorption region was plotted and is shown in Figure 6. The optical band gap energy (E_g) was estimated by extrapolation of the linear portion of the curve to a point $(\alpha h\nu)^2 = 0$. Using this method, the optical energy band gap of the crystal is found as 4.3 eV. This high value of band gap energy and the lower cut-off wavelength of the crystal affirm the suitability of the crystal for photonic and optical applications.

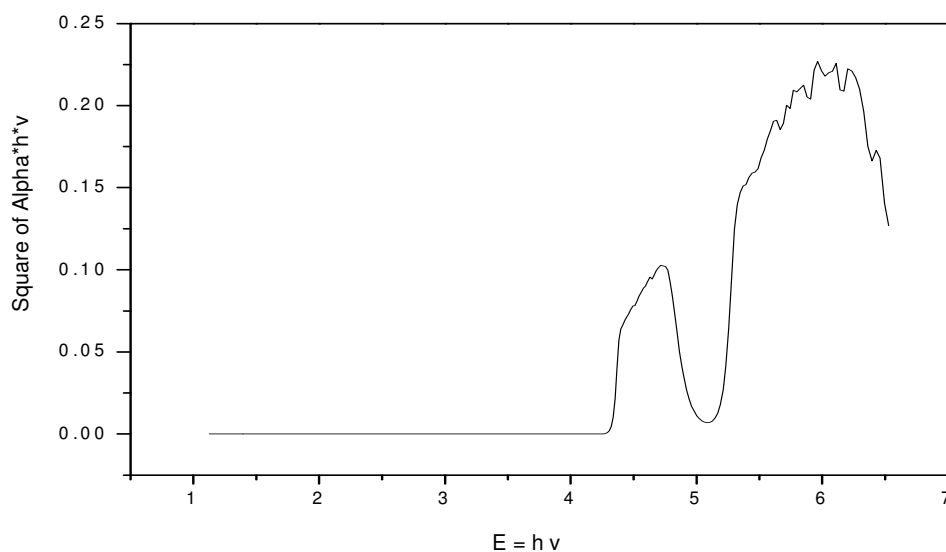


Fig.-6: The variation of $(\alpha h\nu)^2$ and incident photon energy ($h\nu$) of mNA

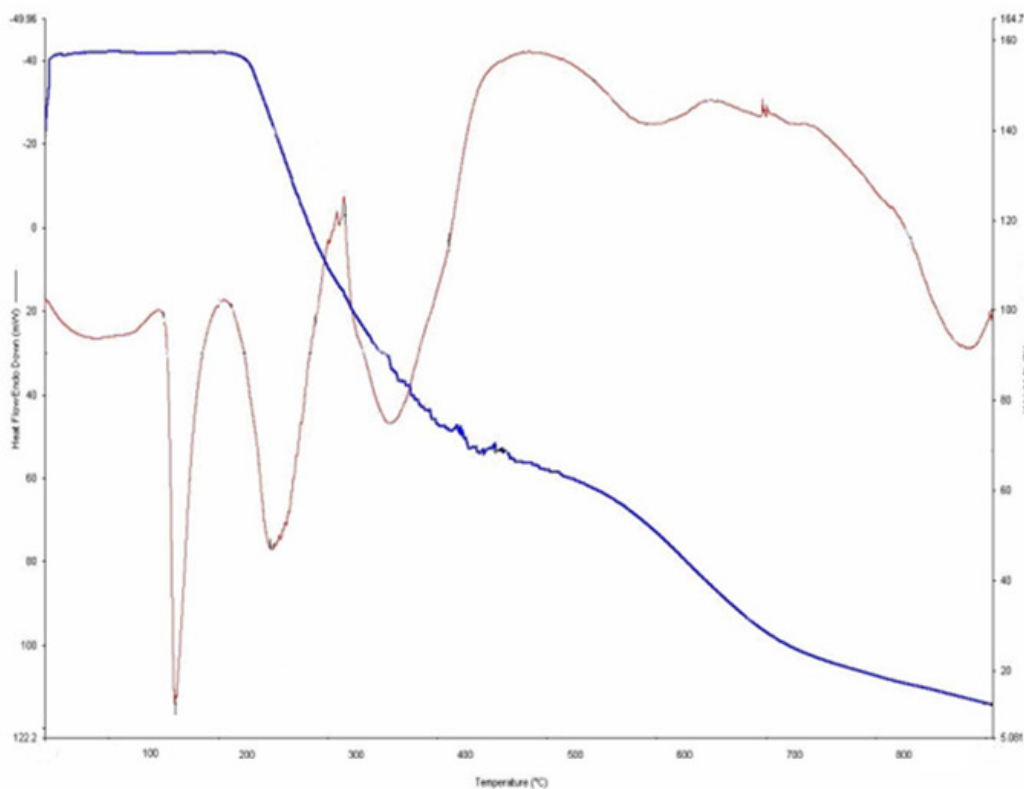


Fig.-7: TG and DTA Thermograms of mNA crystal

Thermal Analysis

The thermal analysis (TGA) study was carried out by using TAQ-500 analyzer at a heating rate of 25⁰ C/min for temperature range 50 to 1000⁰ in nitrogen inert atmosphere to study the weight loss and thermal

stability. The observed thermogram is reported in Figure 7. The crystal was stable up to the temperature of 114°C. The DTA curve shows that this compound undertakes a series of endothermic peaks in a wide temperature range (114-1000°C). The most important one appears at about 114°C. It corresponds to a melting point. From 114°C, the DTA curve shows a series of weak endothermic peaks characterized by an important weight loss observed on the TGA curve²¹. This thermal phenomenon is attributed to the degradation of the organic entity, well confirmed by the consistent residue of the crystal at the end of the experiment. The sharpness of the endothermic peak shows a good degree of crystallinity of the grown crystal²². Thus the grown crystal is a best suitable material for NLO application.

Vickers Hardness Test

Vickers microhardness indentation test is used to characterize the hardness of the material. The hardness number can be evaluated by the knowledge of the load applied and the cross-sectional area of the depth of the impression. Smooth surfaces of as-grown crystals were chosen for the investigation. The Vicker's hardness value is calculated from the formula:

$$H_v = 1.8544 * (P/d^2) \text{ Kg/mm}^2$$

Where P is the applied load in kg and d is the average diagonal length in millimeters of the indented impressions.

In the present study, hardness was measured using Leitz-Wetzler hardness tester. Different loads were applied at a time of 10 seconds. The indentation marks were obtained and measured in terms of the diagonal length (d). Six trails were performed at each load to minimize the error. The plot of load (P) against Vicker's hardness (H_v) is shown in Fig.-8. It is observed that the microhardness increases with the increase of load at lower values which can be attributed to the work hardening of the surface layers.

At higher loads beyond 100 gms, the microhardness shows a tendency to saturate^{23, 24}. Significant cracking occurs which may be due to the release of internal stresses generated locally by indentation. The relation connecting the applied load (P) and diagonal length (d) of the indenter is given by the Meyer Law. From Meyer's Law $P=ad^n$ connecting the applied load (P) and diagonal length (d) of the indentation, the work hardening coefficient 'n' was calculated. Here 'a' is a constant for a given material.

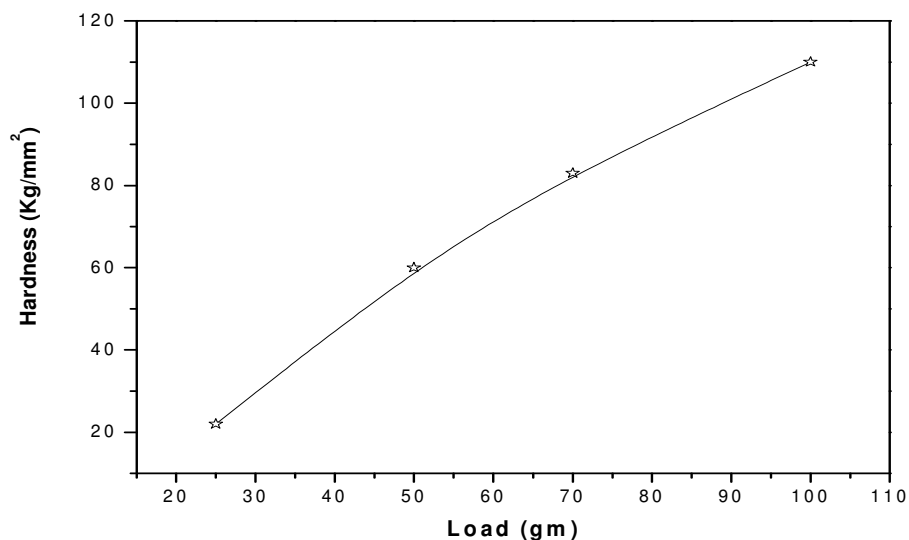


Fig.-8: Vickers Hardness Vs load for mNA Crystals

Elastic Stiffness Constant

The elastic stiffness constant (C_{11}) was calculated for different loads using Wooster's empirical formula²⁵ $C_{11} = H_v^{7/4}$. The C_{11} values are shown in Table.2. These values give an idea about the tightness of bonding between neighboring atoms. The high value shows that the binding forces between the atoms were quite

strong. It was calculated for the loads from 25gm to 100gm. The Plot of H_v versus Elastic Stiffness Constant for the title crystal is shown in Fig.-9.

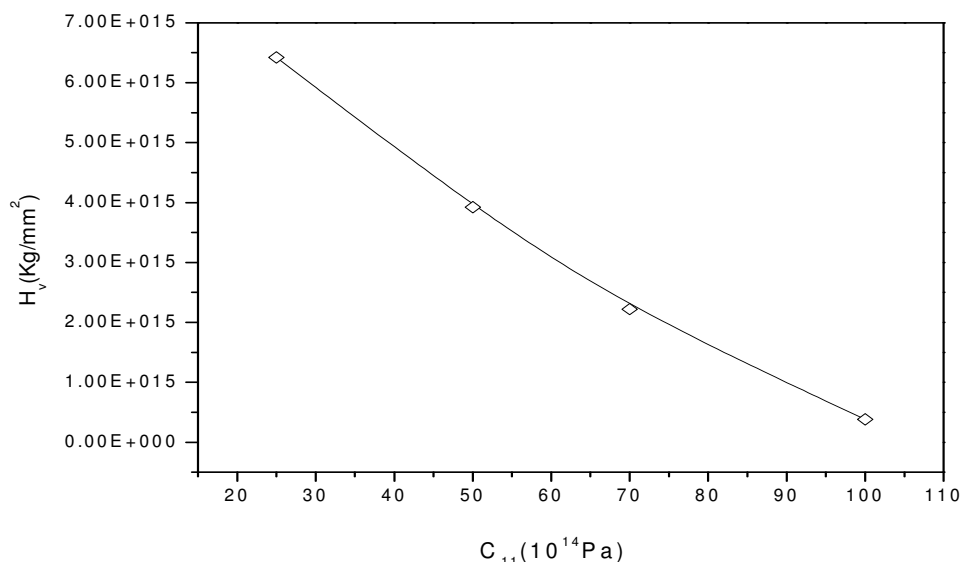


Fig.-9: Plot of H_v versus Load of Elastic Stiffness Constant

Table-2: Elastic Stiffness Constant

Load (g)	C_{11} (10^{14} Pa)
25	64.2093
50	39.2236
70	22.2293
100	3.8406

Dielectric Studies

The frequency dependent measurements of capacitance, C , and $\tan d$, were obtained using a computer controlled LCR meter N4L Numetric QPSM 1735 instrument. The dielectric constant of a material is generally composed of four types of contributions, viz. ionic, electronic, orientational and space charge polarization. All of these may be active at low frequencies. The nature of variations of dielectric constant with frequency and temperature indicates the type of contributions that are present in them. The dipolar orientational effect can be seen in some materials at high frequencies and ionic and electronic polarizations below 10^3 Hz. The large value of ϵ_r at low frequency and at low temperature is due to the presence of space charge polarization, which depends on the purity and perfection of the sample²⁶. Figure-10 shows the variation of dielectric constant with frequency measured at room temperature for the mNA. The dielectric constant is a maximum at low frequency and decreases with increasing of frequency for the crystals. The increase in the dielectric constant at low frequency is attributed to space-charge polarization.

SEM Studies

Chemical analysis and morphological studies were carried out using scanning electron microscopy (SEM Model JSM 840A)²⁷. Figure-11 depicts the SEM image of the crystal. It shows some darker and brighter uneven areas. This might be due to solvent inclusions, which is most commonly observed in solution growth. Interesting features of surface morphologies are observed in SEM, actually exhibits stepped structure. Variations in step directions and the appearance of wider steps could be attributed to the general roughness/grain boundary.

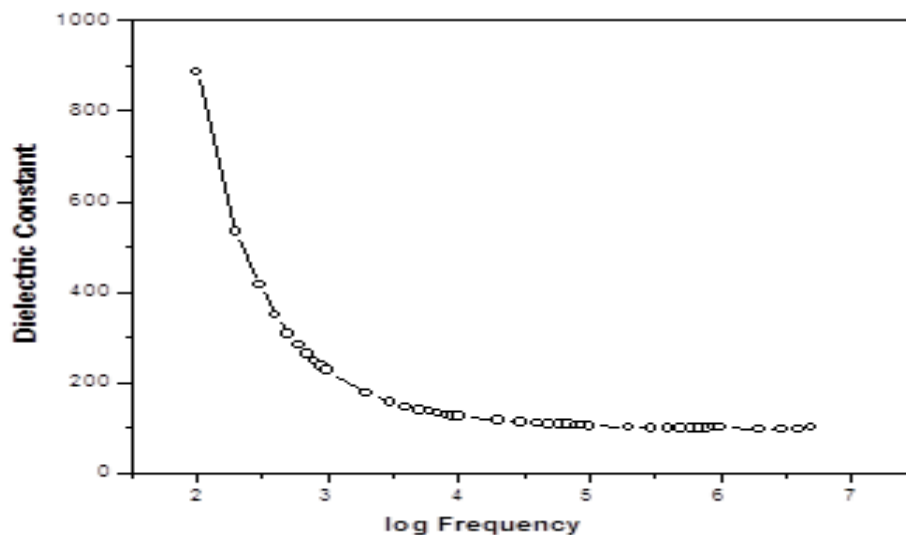


Fig.-10: Variation of Dielectric Constant with Frequency for mNA

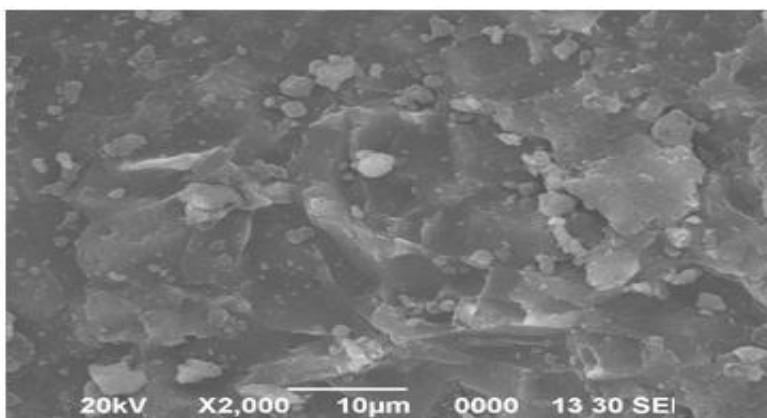


Fig.-11: The SEM Image of the Crystal

SHG Efficiency Measurement

The grown crystals were subjected to the NLO study to measure the efficiency with respect to Meta Nitroaniline. The SHG property of a grown crystal was tested by the Kurtz and Perry powder method. The optical signal generated from the sample was converted into an electrical signal and was measured using an oscilloscope. The measured outputs for Meta Nitroaniline were tabulated in Table-3.

Table-3: SHG efficiency measurement

Input power (mW)	Output power (mW)		SHG conversion efficiency (%)	
	KDP	mNA	KDP	mNA
285	190	210	67.0	73.0
468	230	250	49.0	53.0
688	320	320	47.0	52.0
830	370	385	41.0	46.0

FTIR Measurements

The room temperature Fourier transform infrared spectrum of mNA is shown in Fig.-12 and it was recorded in the region $400\text{-}4000\text{cm}^{-1}$ at a resolution of 4cm^{-1} using Perkin Elmer Fourier transform Infrared Spectrophotometer, model SPECTRUM RX1, using KBr pellets containing a fine mNA powder obtained from the grown single crystals, equipped with a LiTaO_3 detector, a KBr beam splitter, He-Ne

Laser source and boxcar apodization used for 250 averaged interferograms collected for both the sample and the background.

The present vibrational spectroscopic study was carried out with a view to obtaining an insight into the structural aspects of optical nonlinearity of crystals^{28, 29}. In order to understand the existence of bonding nature of the materials, the present study has been undertaken. Thus the molecular structure of the synthesized compound was confirmed by the spectral analysis. The observed wave numbers, relative intensities obtained from the recorded spectra and the assignments proposed for the title nonlinear optical crystal is given in Table-4. The assignments of bands observed in the vibrational spectrum are an essential step for solving the structural and chemical problem.

Table-4: The bands observed in FTIR and its assignments

FTIR Wavelength	Assignments
3778	NH ₂ – Asymmetric Bending
3407	NH ₂ – Asymmetric Bending
3124	NH ₂ – Asymmetric Bending
3032	NH ₂ – Asymmetric Bending
2920	NH ₂ –symmetric Stretching
2819	NH ₂ –symmetric Stretching
2701	NH ₂ – Asymmetric Bending
2612	NH ₂ – Asymmetric Bending
2276	C-H Stretching
1950	C-C Stretching
1825	C-N Asymmetric Stretching
1594	C=C aromatic stretching
1486	C-N Asymmetric Stretching
1444	C-N Asymmetric Stretching
1324	CN stretching
1179	Ring in-plane bending, CH in-plane bending
1101	CH in-plane bending
1055	Ring in-plane bending, C–O stretching
933	C–C stretching
892	C-H out of plane bending
836	C-H Ring Puckering
756	C-N symmetric Stretching
657	Ring in-plane bending, C–C Stretching
615	C-H deformation

Photoluminescence Studies

Photoluminescence (PL) measurement of mNA is taken at room temperature with 20 mW, 633 nm He–Ne laser is used for the excitation energy and shown in Fig.-13. A strong emission peak is observed from the spectrum at 355 nm. This reveals that the grown crystal has a green emission property. This reveals that the grown crystal has a green emission property and also this strong peak indicates there may be the presence of intrinsic defects in the forbidden band region³⁰.

Hyper Polarizability Calculation

For calculating the hyperpolarizability, the optimization has been carried out in the unrestricted open-shell Hartree-Fock level. The geometries are fully optimized without any constraint with the help of analytical gradient procedure implemented within Gaussian 03W program³¹. The electric dipole moment and dispersion-free first-order hyperpolarizability are calculated using finite field method. The finite field method offers a straightforward approach to the calculation of hyperpolarizability³². The 3-21(d,p) basis

set gives remarkably good geometries for such a small basis set and in fact it is used for the geometry optimization of some high accuracy energy methods.

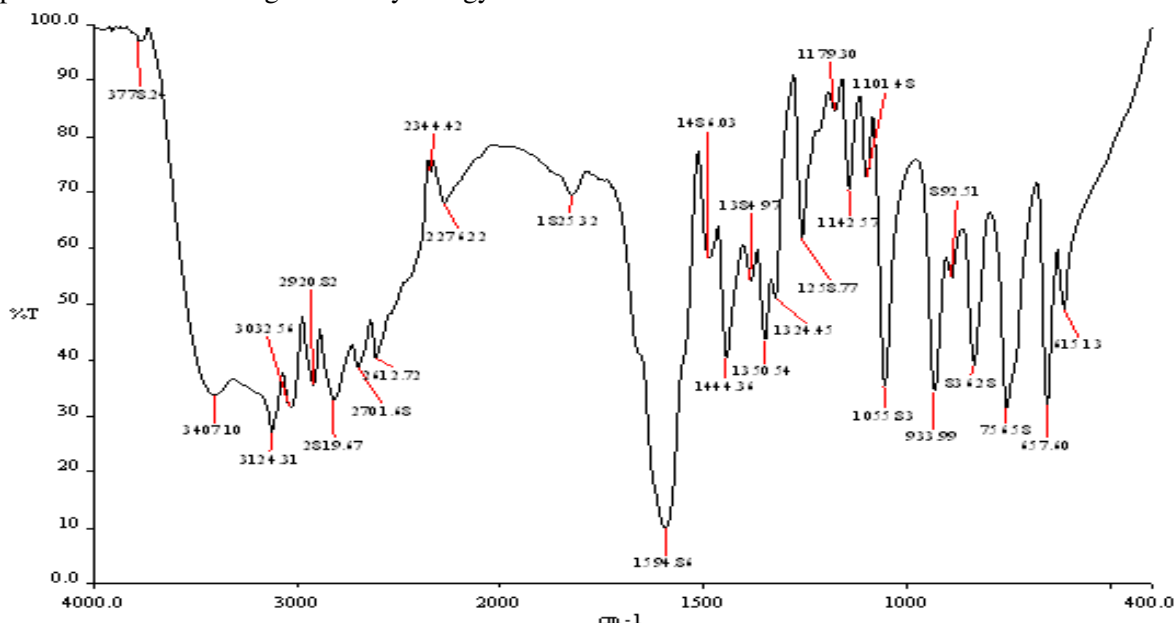


Fig.-12: FTIR spectrum of mNA

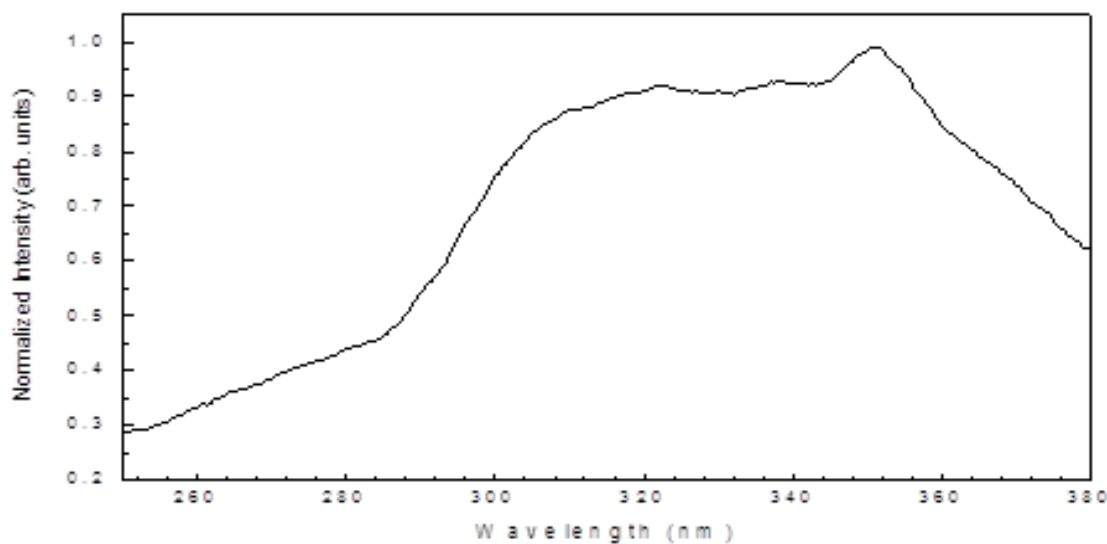


Fig.-13: Photoluminescence Spectrum of Growth mNa Crystal

The nonlinear properties of an isolated molecule in an electric field $E_i(\omega)$ can be represented by the Taylor expansion of the total dipole moment m_i induced by the field is taken at zero fields, Dipole moment:

$$\mu_i = - \left[\frac{\partial^2 E}{\partial F_i \partial F_j} \right]_0$$

Components of polarizability tensor:

$$\alpha_{ij} = - \left[\frac{\partial E}{\partial F_i} \right]_0$$

Components of hyperpolarizability tensor:

$$\beta_{ijk} = - \left[\frac{\partial^3 E}{\partial F_i \partial F_j \partial F_k} \right]_0$$

These components are to be distorted by an external electric field. The value of total static polarizability and hyperpolarizability are obtained from the following equation,

$$\beta_{tot} = [(\beta_{xxx} + \beta_{xyy} + \beta_{xzz})^2 + (\beta_{yyy} + \beta_{yzz} + \beta_{yxx})^2 + (\beta_{zzz} + \beta_{zxx} + \beta_{zyy})^2]^{1/2}$$

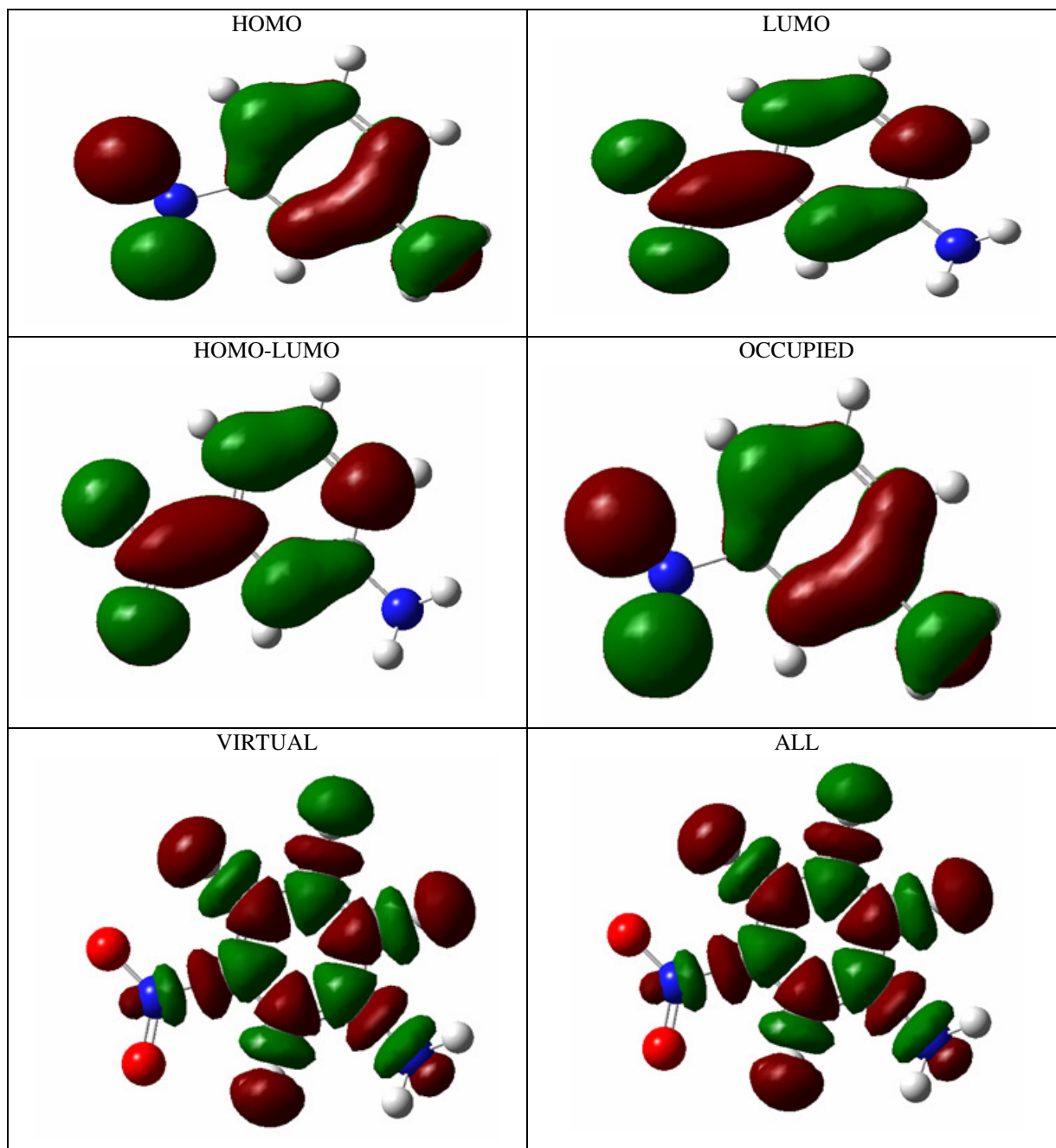


Fig.-14: Molecule orbital of Meta Nitroaniline

In the presence of an applied electric field, first-order hyperpolarizability is a third rank tensor that can be described by a 3×3×3 matrix. The components of the 3D matrix can be reduced to 10 components

because of the Kleinman symmetry³³. The matrix can be given in the lower tetrahedral format. It is obvious that the lower part of the 3×3×3 matrix is tetrahedral. The calculation of NLO properties with high accuracy is challenging and requires consideration of many different issues. Computational techniques are becoming valuable in designing, modeling and screening novel NLO materials. The calculated value of hyperpolarizability for the title compound is 0.402499×10^{-30} esu, which is nearly 2 times that of urea (0.1947×10^{-30} esu). The calculated values of dipole moment and hyperpolarizability values are tabulated in Table-5. The β_{zxx} direction shows the biggest value of hyperpolarizability which insists that the delocalization of electron cloud is more that direction than other directions.

Figure-14 shows the highest occupied molecule orbital (HOMO) and lowest unoccupied molecule orbital (LUMO) of Meta Nitroaniline. There is an inverse relationship between hyperpolarizability and HOMO–LUMO.

$$\begin{aligned} \text{HOMO energy} &= -0.266 \text{ a.u} \\ \text{LUMO energy} &= 0.169 \text{ a.u} \\ \text{HOMO–LUMO energy gap} &= 0.435 \text{ a.u} \end{aligned}$$

Table-5: The dipole moment (μ) and first-order hyperpolarizability (β) of mNA derived from DFT calculations

β_{xxx}	0.46927
β_{xxy}	-0.11436
β_{xyy}	-0.27422
β_{yyy}	-0.00713
β_{zxx}	0.205357
β_{xyz}	0.072191
β_{zyy}	-0.15792
β_{xzz}	0.155044
β_{yzz}	0.129833
β_{zzz}	0.15098
β_{total}	0.402499
μ_x	0.02525878
μ_y	6.1127E-05
μ_z	0.00874343
μ	0.18456254

Dipole moment (μ) in Debye, hyperpolarizability $\beta(-2\omega; \omega, \omega) 10^{-30}$ esu.

CONCLUSION

The good optical quality Meta Nitroaniline (mNA) single crystals were grown by solution growth method at room temperature. The lattice parameters were found by single crystal XRD technique. From the UV-Vis spectrum, it is seen that the lower cut off wavelength of mNA were around 280 nm and its transparency in the entire visible region shows that this material is a best suitable candidate for optoelectronic application. The optical energy band gap of the crystal is found as 4.3 eV and this asserts the suitability of the crystal for photonic and optical applications. The DTA curve shows that this compound undertakes a series of endothermic peaks in a wide temperature range (114-1000°C). The mechanical studies indicate that the microhardness increases with the increase of load at lower values which can be attributed to the work hardening of the surface layers. The elastic stiffness constant (C_{11}) was calculated for different loads using Wooster's empirical formula. From Dielectric studies, the dielectric constant is a maximum at low frequency and decreases with increasing of frequency for the crystals. The vibrational spectroscopic study was carried out with a view to obtaining an insight into the structural aspects of optical nonlinearity of grown crystal. The photoluminescence study reveals that the grown crystal has a green emission property. The calculated hyperpolarizability value found to be nearly 2 times that of urea.

REFERENCE

1. V. Krishna Kumar, R. Nagalakshmi, *Spectrochim. Acta A. Mol. Biomol. Spectrosc.* **66**, 924 (2007), DOI: [10.1016/j.saa.2006.03.042](https://doi.org/10.1016/j.saa.2006.03.042)

2. P. Baskaran, M. Vimalan, P. Anandan, G. Bakiyaraj, K. Kirubavathi, S. G. Praveen and K. Selvaraju, *Materials Research Express*, **3**, 035101 (2016), DOI: [10.1088/2053-1591/3/3/035101](https://doi.org/10.1088/2053-1591/3/3/035101)
3. Viji niraimathi, V. Aroulmoji and G. Rajarajan, *American Journal of Materials Science and Engineering*, **2**, 36 (2014), DOI: [10.12691/ajmse-2-3-2](https://doi.org/10.12691/ajmse-2-3-2)
4. T. Jayanalina, G. Rajarajan, K. Boopathi and K. Sreevani, *Journal of Crystal Growth*, **426**, 9 (2015), DOI: [10.1016/j.jcrysgro.2015.05.014](https://doi.org/10.1016/j.jcrysgro.2015.05.014)
5. N. Senthilvelan, G. Rajarajan, S. Sivakumar, J. Elanchenzhiyan and A. Jegatheesan, *Rasayan Journal of Chemistry*, **10**, 245 (2017), DOI: [10.7324/RJC.2017.1011566](https://doi.org/10.7324/RJC.2017.1011566)
6. P. Kumaresan, S. Moorthy Babu and P.M. Anbarasan, *Optic. Mater.*, **30**, 1361 (2008), DOI: [10.1016/j.optmat.2007.07.002](https://doi.org/10.1016/j.optmat.2007.07.002)
7. N. Senthilvelan, G. Rajarajan, A. Jegatheesan, S. Sivakumar and J. Elanchenzhiyan, *Rasayan Journal of Chemistry*, **10**, 218 (2017), DOI: [10.7324/RJC.2017.1011565](https://doi.org/10.7324/RJC.2017.1011565)
8. J. Zyss, R. Masse, M. Bagieu-Beucher, J. P. Levy, *Adv. Mater.*, **5**, 120 (1993), DOI: [10.1002/adma.19930050210](https://doi.org/10.1002/adma.19930050210)
9. Y. Le Fur, M. Bagieu-Beucher, R. Masse, J.-F. Nicoud, J.-P. Levy, *Chem. Mater.*, **8**, 68 (1996), DOI: [10.1002/adma.19930050210](https://doi.org/10.1002/adma.19930050210)
10. G.F. Lipscomb, A.F. Garito, R.S. Narang, *J. Chem. Phys.*, **75**, 1509 (1981), DOI: [10.1063/1.442157](https://doi.org/10.1063/1.442157)
11. B.F. Levine, C.G. Bethea, C.D. Thurmond, R.T. Lynch, J.L. Bernstein, *J. Appl. Phys.*, **50**, 2523 (1979), DOI: [10.1063/1.326248](https://doi.org/10.1063/1.326248)
12. J. Hernandez-Paredes, O. Hernandez-Negrete, R.C. Carrillo-Torres, R. Sanchez-Zeferino, A. Duarte-Moller, M.E. Alvarez-Ramos, *Spectrochim. Acta, Part A*, **149**, 240 (2015), DOI: [10.1016/j.saa.2015.04.080](https://doi.org/10.1016/j.saa.2015.04.080)
13. S.P. Karna, P.N. Prasad and M. Dupuis, *J. Chem. Phys.*, **94**, 1171 (1991), DOI: [10.1063/1.460024](https://doi.org/10.1063/1.460024)
14. A. Dey and G.R. Desiraju, *Chem. Commun.*, **19**, 2486 (2005), DOI: [10.1039/B502516H](https://doi.org/10.1039/B502516H)
15. J. Hernandez-Paredes, A.L. Olvera-Tapia, J.I. Arenas-García, H. Höpfl, H. Morales-Rojas, D. Herrera-Ruiz, A.I. Gonzaga-Morales, L. Rodríguez-Fragoso, *Crystal Engineering Communication*, **17**, 5166 (2015), DOI: [10.1039/C4CE01934B](https://doi.org/10.1039/C4CE01934B)
16. T.S. Thakur, R. Dubey, G.R. Desiraju, *IUCrJ*, **2**, 159 (2015), DOI: [10.1107/S205225251500189X](https://doi.org/10.1107/S205225251500189X)
17. D.Y. Curtin, I.C. Paul, *Chem. Rev.*, **81**, 525 (1981), DOI: [10.1021/cr00046a001](https://doi.org/10.1021/cr00046a001)
18. V. Arjunan, Mariusz K. Marchewka, A. Pietraszko, M. Kalaivani, *Spectrochimica Acta Part A: Molecular and Biomolecular Spectroscopy*, **97**, 625 (2012), DOI: [10.1016/j.saa.2012.07.018](https://doi.org/10.1016/j.saa.2012.07.018)
19. A. Jonie Varjula, C. Vesta, C. Justin Raj, S. Dinakaran, A. Ramanand, S. Jerome Das, *Materials Letters*, **61**, 5053 (2007), DOI: [10.1016/j.matlet.2007.04.012](https://doi.org/10.1016/j.matlet.2007.04.012)
20. B. Neelakantaprasad, A. Jegatheesan, J. Murugan, V. Aroulmoji and G. Rajarajan, *Asian Journal of Chemistry*, **26**, 176 (2014), DOI: [10.14233/ajchem.2014.15368](https://doi.org/10.14233/ajchem.2014.15368)
21. NP. Rajesh, V. Kannan, M. Ashok, K. Sivaji, P. Santhana Raghavan and P. Ramasamy, *Journal of Crystal Growth*, **262**, 561 (2004), DOI: [10.1016/j.jcrysgro.2003.10.064](https://doi.org/10.1016/j.jcrysgro.2003.10.064)
22. T Jayanalina, G Rajarajan, S Parthiban, and SC Mojumdar. *Journal of Thermal Analysis and Calorimetry*, **112**, 1025 (2013), DOI: [10.1007/s10973-013-3058-7](https://doi.org/10.1007/s10973-013-3058-7)
23. T. Kanagasekaran, P. Mythili, P. Srinivasan, N. Shailesh Sharma, R. Gopalakrishnan, *Materials Letters*, **62**, 2486 (2008), DOI: [10.1016/j.matlet.2007.12.039](https://doi.org/10.1016/j.matlet.2007.12.039)
24. A. Jegatheesan, G. Rajarajan, B. Ravi and B. Neelakantaprasad, *Journal of Chemical and Pharmaceutical Research*, **7**, 2173 (2015)
25. W.A. Wooster, *Rep. Phys. Phys.*, **16**, 62 (1953), DOI: [10.1088/0034-4885/16/1/302](https://doi.org/10.1088/0034-4885/16/1/302)
26. M. Narayan Bhat and SM Dharamaprakash, *Journal of Crystal Growth*, **242**, 245 (2004), DOI: [10.1016/S0022-0248\(02\)01327-1](https://doi.org/10.1016/S0022-0248(02)01327-1)
27. R. Ezhil Vizhi and S. Kalainathan, *Materials Letters*, **62**, 791 (2008), DOI: [10.1016/j.matlet.2007.06.083](https://doi.org/10.1016/j.matlet.2007.06.083)
28. A Jegatheesan, J Murugan, B Neelagantaprasad and G Rajarajan, *International Journal of Computer Applications* **53**, 15 (2012), DOI: [10.5120/8408-2040](https://doi.org/10.5120/8408-2040)

29. G. Senthil Murugan, N. Balamurugan and P. Ramasamy, *Materials Letters*, **62**, 3087 (2008), DOI: [10.1016/j.matlet.2008.01.137](https://doi.org/10.1016/j.matlet.2008.01.137)
30. SG. Bhat and SM. Dharaamaparakash, *Materials Research Bulletin*, **33**, 833 (1998), DOI: [10.1016/S0025-5408\(98\)00049-X](https://doi.org/10.1016/S0025-5408(98)00049-X)
31. M.J. Frisch, G.W. Trucks, H.B. Schlegel, G.E. Scuseria, M.A. Robb, J.R. Cheeseman, J.A. Montgomery, Jr., T. Vreven, K.N. Kudin, J.C. Burant, J.M. Millam, S.S. Iyengar, J. Tomasi, V. Barone, B. Mennucci, M. Cossi, G. Scalmani, N. Rega, G.A. Petersson, H. Nakatsuji, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, M. Klene, X. Li, J.E. Knox, H.P. Hratchian, J.B. Cross, C. Adamo, J. Jaramillo, R. Gomperts, R.E. Stratmann, O. Yazyev, A.J. Austin, R. Cammi, C. Pomelli, J.W. Ochterski, P.Y. Ayala, K. Morokuma, G.A. Voth, P. Salvador, J.J. Dannenberg, V.G. Zakrzewski, S. Dapprich, A.D. Daniels, M.C. Strain, O. Farkas, D.K. Malick, A.D. Rabuck, K. Raghavachari, J.B. Foresman, J.V. Ortiz, Q. Cui, A.G. Baboul, S. Clifford, J. Cioslowski, B.B. Stefanov, G. Liu, A. Liashenko, P. Piskorz, I. Komaromi, R.L. Martin, D.J. Fox, T. Keith, M.A. Al-Laham, C.Y. Peng, A. Nanayakkara, M. Challacombe, P.M.W.G. ill, B. Johnson, W. Chen, M. W. Wong, C. Gonzalez, J.A. Pople, Gaussian Inc, Wallingford CT, (2004)
32. A. Sudharsan, S. Seshadri, T. Gnanasambandan, R.R. Saravanan, *Spectrochim.Acta A Mol. Biomol. Spectrosc.*, **131**, 432 (2014), DOI: [10.1016/j.saa.2014.04.098](https://doi.org/10.1016/j.saa.2014.04.098)
33. D.A. Kleinman, *Physical. Review*, **126**, 1977 (1962)

[RJC-1655/2018]